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Soft nanocomposites: nanoparticles to tune gel properties

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Review

Soft nanocomposites: nanoparticles to tune gel properties

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Keywords: gel nanocomposites, nanoparticles, hydrogels, stimuli-responsiveness

Abstract

The demand for new, soft materials with bespoke physical and biological characteristics and functionality has fuelled the research into nanocomposite hydrogels. 'Soft' nanocomposites – nanoparticles within a hydrated, polymeric gel matrix – offer a simple, yet versatile, platform for the design of materials with specific – and tunable – properties. Indeed, the 'soft' properties of the matrix can be combined with the inherent functionality of the nanoparticles (drug loading, antimicrobial, light refraction...) or give rise to altogether new characteristics (toughness, optical properties, self-healing...), evolved from the synergistic interaction of the polymer chains with the particles. In this review, we report the evolution and achievements of nanocomposite gels, with a focus on mechanisms and structure. The review therefore is structured around the *properties* resulting from the gel/nanoparticles association – rather than a classification based on applications or specific types of polymer or nanoparticles. *How* can nanoparticles tune mechanical, optical, biological properties or impart stimuli-responsiveness to a polymer gel matrix - and how is this behaviour linked to the underlying structure?

Abbreviations: NC (nanocomposite); NP (nanoparticle); poly(ethylene oxide) (PEO); poly(propylene oxide) (PPO); N-isopropyl acrylamide (NIPAAm); poly(*N*-isopropyl acrylamide) (PNIPAAm); polyacrylamide (PAAm); poly(*N,N*-dimethylacrylamide) (PDMA); crystalline colloidal arrays (CCA); near-infra red (nIR); Small-angle neutron scattering (SANS)

1. Introduction

This review is about the marriage between two areas of soft matter that both separately have experienced a tremendous growth over the past couple of decades: hydrogels and nanotechnology. Both have been stimulated by the demand for new materials with very specific characteristics for a plethora of applications – as sensors, actuators, for drug delivery, tissue repair etc – as well as the simple fascination for building new structures, functions and properties. While hydrogels have been known and made for a very long time, the more recent surge in hydrogel research has been largely driven by their similarity with bodily tissues (thus offering a favourable environment to grow cells and deliver active compounds) as well as their versatile nature, making them unique candidates to impart responsiveness to specific triggers in so-called ‘smart’ materials,¹⁻³ while nanotechnology also encompasses a vast range of rapidly expanding sectors, from microelectronics, catalysis to diagnostic and drug delivery.⁴⁻¹⁰ The combination of hydrogels with nanoparticles therefore opens up enormous opportunities for creating new materials and properties, which may simply arise from the added properties of the bulk and the dispersed phase, but also from a synergistic interaction between the two components, leading to totally new characteristics (**Table 1**).

Table 1. Examples of gel matrix-nanoparticles used to obtain gel nanocomposites and their outcomes

Gel nanocomposites				
Aspect	matrix	NP	effect	Ref
Mechanical	PNIPAAm	inorganic Clays	enhanced resilience and extensiveness	23-25, 27, 28
	PAAm	graphene oxide	enhanced resilience and extensiveness	30, 31
	PAAm	Silica	enhanced tensile modulus	32, 33
	PAC-DMAA	Titania	enhanced resilience and extensiveness	34

	PAAm	layered double hydroxide platelets	enhanced tensile modulus	35
	PDMA	cellulose nanocrystals	enhanced tensile modulus	55
Optical	Gelatin	Silver	colour change	11
	Gelatin	Copper	colour change	60
	Inorganic polymers	Sulfides	refractive index manipulation	66-68
	Inorganic polymers	Titanium oxide	refractive index manipulation	58
	Inorganic polymers	laponite	domains formation	78
	Inorganic polymers	crystalline colloidal arrays	colour changes	71, 72, 82, 83
	Inorganic polymers	inverse opal structures	colour changes	74, 85, 86
Biological	Inorganic polymers	Silver-graphene oxide	Antimicrobial, accelerates healing	16
	collagen and gelatin	Silver	antimicrobial	90, 91
	modified hyaluronic acid	calcium phosphate	bone regeneration	92
	hyaluronan	poly(lactide-co-glycolide)	sustained drug delivery	93
	PNIPAAm-AAm-PEGDA	graphene oxide and iron oxide	controlled drug delivery	94
	Inorganic polymers	few-walled carbon nanotubes	localized mechanical actuation	95
Stimuli-responsiveness	PNIPAAm	clay	temperature	24, 26, 103
	double-network PNIPAAm	silica	improved temperature response	97
	PNIPAAm	PNIPAAm	improved temperature response	98
	PNIPAAm	Iron oxide	magnetic response	6, 104
	P(NIPAAm-co-AAm)	gold nanoshells	optical response	105
	PNIPAAm	gold	optical response	107
	PNIPAAm	SW carbon nanotubes	optical response	108
	PNIPAAm	Graphene oxide	optical response	109
	polypeptide PC10P	gold nanorods	optical response	110
F127	super paramagnetic iron oxide	improved temperature response	113	

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3 Apart from a few historic examples,¹¹ gel nanocomposites are a much younger topic
4 than their elder sibling – and not to be confused with - ‘nanocomposites’, where the
5 matrix is pure polymer. A search on Web of Science database shows that
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10 ‘nanocomposite gels’ only attracted a handful of hits pre-2000s, but publications
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12 have risen steeply in the last 10 years, reaching ca. 1500 papers and > 30,000
13 citations p.a. (**Figure 1**). As with nanocomposites (NC), the most traditional purpose
14 of adding nanoparticles (NPs) into a gel matrix (and still a highly productive area of
15 research) has been to improve mechanical properties. Indeed, it has now been
16
17 successfully demonstrated that the introduction of NPs can open new avenues
18 towards material design, leading to gels which, while containing at least 90% water,
19 can display extraordinary toughness.^{5, 12} An area where both hydrogels and
20 nanoparticles, independently, have generated intense research is in biomedicine. As
21 a result, a plethora of NC gels have been proposed to address challenges in tissue
22 engineering and drug delivery,^{8, 13-15} for instance for antimicrobial applications (with
23 metal NPs),¹⁶ controlled drug release,¹⁷ regenerative medicine⁹ (in particular with
24 ceramic fillers, such as calcium phosphate), carbon-based nanomaterials,¹⁸ and
25 biopolymers as the gel matrix.^{8, 19} Increasing activity has also been focused on
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27 ‘smart’ materials, and we show in this review that NC gels provide an ideal platform
28 to impart responsiveness to soft materials.
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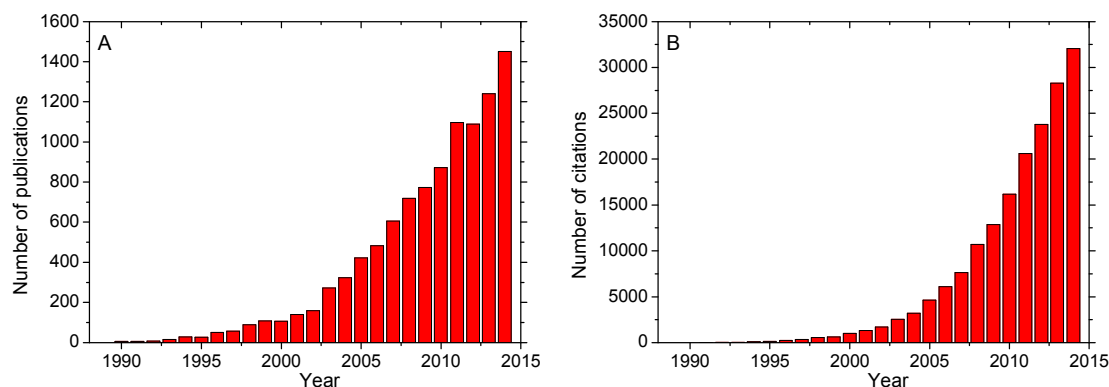


Figure 1. Number of publications (A) and citations (B) containing the search keywords 'nanocomposite' and 'gel' in the last 15 years according to ISI web of science (Sept/2015). It reveals an exponential growth in both number of papers and citations.

This review is structured around *properties* (**Table 1**): how can nanoparticles be used to modulate the characteristics of a gel matrix? Hence our aim is not to give recipes for sophisticated preparation and synthesis protocols, or indeed report on the use of gels as templates for in situ fabrication of NPs, but rather to show how a range of new interesting properties can be built by associating nanoparticles and gels, specifically: mechanical, optical, biological and responsiveness to stimuli. We conclude with a section which summarises some of the – still very few - insights into the structure of NC gels underlying these unique properties.

2. Mechanical properties

The first and still most widespread intention when introducing nanoparticles in a gel matrix is the improvement of mechanical properties. Traditional chemically crosslinked gels have a number of limitations, due to the intrinsically random crosslinking process, resulting in morphological inhomogeneity (opacity) and poor mechanical performance (brittleness, limited elongation, lack of toughness etc). Compared to other types of strategies towards the design of strong hydrogels (such as interpenetrated networks,²⁰ sliding gels,²¹ tetraPEG²²), hydrogels made by combining polymer chains with inorganic fillers offer a relatively simple answer,

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3 added to the possibility of easily tuning properties by varying composition. The
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5 discovery of gels with high toughness and extensibility has revived interest in these
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7 materials, and led to applying to the field of hydrogels techniques and theories
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9 previously developed for rubbers.
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11 Haraguchi²³⁻²⁶ was the first to propose solving the conundrum of controlling both the
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13 density of crosslinks (ν) and inter-crosslinks molecular weight. This was achieved by
14
15 designing a nanocomposite hydrogel where the polymerisation of *N*-alkylacrylamide
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17 derivatives (such as *N*-isopropyl acrylamide, NIPAAm) was initiated from water-
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19 swellable inorganic clay platelets (**Figure 2 A**).²³ This novel class of NC hydrogels,
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21 with high transparency, results in remarkable properties, in particular toughness,
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23 capacity to withstand high levels of deformation, high elongation^{23, 27} (ca. 1500%)
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25 with near-complete recovery and self-healing (**Figure 2 B,C**).²⁸ In these materials,
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27 neighbouring clay platelets are connected by polymer chains so that they act as
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29 high-functionality crosslink points; therefore, clay-clay interparticle distance is
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31 equivalent to inter-crosslink distance and thus controlled (**Figure 2 A**). The nature of
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33 the grafting points on the clay surface is attributed to a combination of ionic
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35 interactions between an anionic end of the polymer chain bonded to SO_3^- and K^+
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37 (from the potassium peroxydisulfate initiator) on the platelets, and coordination
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39 interaction between nucleophilic N(H)CO from PNIPAAm and Si from the clay
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41 surface. Since the polymer chains are not restricted by the presence of chemical
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43 crosslinks, they behave like free, linear polymers, resulting in very high swelling
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45 ratios ($W_w/W_{dry} = 110$)²³ and rapid deswelling with temperature changes. The
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47 structure of these gels, at rest and under elongation, obtained from small-angle
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49 neutron and X-ray scattering, is discussed further in section 6 (and **Figure 7**), which
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51 is dedicated to NC gels architecture.
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3 Following the work of Haraguchi, many other nanocomposite gels have been
4 developed (**Figure 2**), using the same strategy of in-situ polymerisation from layered
5 inorganic nanoparticles, either with clay,²⁹ graphene oxide,^{30, 31} silica,^{32, 33} titania³⁴ or
6 layered double hydroxide platelets (LDH) (**Figure 2 E**),³⁵ all showing remarkable
7 mechanical toughness, elongation and self-healing properties. The
8 LDH/polyacrylamide (PAAm) hydrogels developed by Hu et al³⁵ show elongations at
9 break > 4000% at low LDH content (2.3%) with an unexpected yielding phenomenon
10 upon stretching, attributed to an unusual hierarchical porous morphology with
11 interconnected pores; the swelled hydrogels display increased tensibility (> 6236%,
12 beyond measuring range), while yielding disappeared. The enhanced mechanical
13 properties are again attributed to a highly dense network of chains crosslinked by
14 NPs, where the polymer retains a high flexibility and mobility between multi-
15 functional crosslink points (**Figure 2**).

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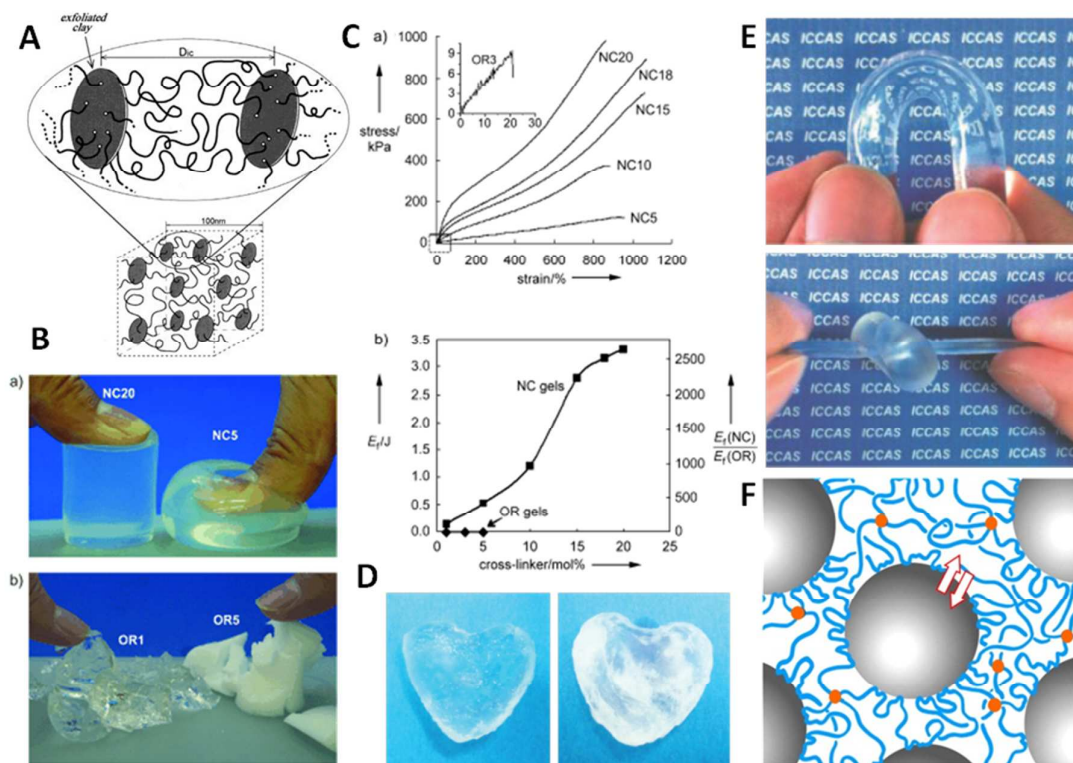


Figure 2. Examples of nanocomposite (NC) gels presenting remarkable mechanical properties. **A.** Organic/inorganic networks developed by Haraguchi, consisting of uniformly dispersed (exfoliated) inorganic clay sheets with PNIPAAm (reprinted with permission from reference 26. Copyright 2002 American Chemical Society). **B.** The resulting NC gels do not rupture under repeated compression by hand, compared to conventional chemically crosslinked gels of PNIPAAm (OR, bottom picture) (reprinted from reference 24 with permission from John Wiley & Sons, Inc.). **C.** a) Tensile stress versus strain curves with increasing amounts of clay (NC5-NC20) show that initial modulus of elasticity (E_i) and tensile strength (σ) increase monotonically with increasing clay content. NC20 displays a $\sigma \sim 1000$ kPa, $E_i \sim 400$ kPa and ca. 1000% elongation at break (inset: magnified view of equivalent OR gel, which is very brittle). b) The fracture energy E_f of NC20 is 2650-times higher than OR gels (reprinted from reference 24 with permission from John Wiley & Sons, Inc.). **D.** NC gels based on clay and dendritic binders retain their shape after being moulded into a heart shape, even after full exchange of water with THF (reprinted by permission from Macmillan Publishers Ltd from reference 36, copyright 2010) **E.** NC gels based on layered double hydroxide platelets (LDH)/PAAm also exhibit remarkable mechanical toughness – here shown under torsion or knotting (reprinted from reference 35 with permission from John Wiley & Sons, Inc.). **F.** Physical adsorption can also be used to create strong NC gels, here with covalently crosslinked (orange) PDMA on silica NPs. The chemical network controls

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3 strain recovery through the entropic restoring forces and silica NPs promote transient and
4 recoverable connectivity (reprinted with permission from reference 37. Copyright 2013
5 American Chemical Society).
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9 Xie et al developed poly(acrylic acid) hydrogels from silica NPs, where the elasticity
10 arises from the entanglements of the polymeric chains and the silica NPs play the
11 role of 'analogous crosslinking points'.³⁸ The tensile strength and elongation at break
12 of the NC gels could be modulated (up to 313 kPa and 3420%) by varying the mass
13 fraction and/or diameters of the vinyl hybrid silica NPs (VSNP), which transfer stress
14 to the polymeric chains.³⁹ When a NC gel is subjected to stress, the intermolecular
15 hydrogen bonds dynamically break and recombine to dissipate energy, resulting in a
16 reorganisation of the polymer chains, while the VSNP maintain the gel network and
17 tolerate stress, even when part of the intermolecular hydrogen bonds start to break.
18 After gel network homogenization, the applied stress can be rapidly and uniformly
19 distributed over the entire network with the multifunctional VSNP acting as transfer
20 centres, thus explaining the remarkable mechanical performance.
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35 Recently, the group of Aida^{36, 40} has developed NC gels that can be moulded into
36 shape-persistent, free-standing objects, with high mechanical strength and rapidly
37 self-heal (**Figure 2 D**). These gels are either based on mixtures of clay nanosheets,
38 a dendritic molecule based on PEO, and sodium polyacrylate³⁶ or photocatalytic
39 titania nanosheets with NIPAAm (**Figure 5 A**), which becomes polymerised *in situ*
40 upon exposure to light,⁴⁰ leading to modulable hydrogels with photolatently reactive
41 crosslinking points.
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50 The NC gels described above all show superior mechanical properties, however they
51 do require a non-trivial level of preparation, and in particular in-situ polymerisation in
52 the presence of a suspension of inorganic particles. Some effort therefore has also
53 been concentrated in developing NC gels from 'off-the-shelf' materials or simpler
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3 materials, in particular through mixing of polymer and particles, where the
4 reinforcement originates from *physical* interactions between the polymer and
5 particles (often clay or silica), where the polymer chains are either non or lightly
6 crosslinked, or possibly exist as a pre-existing physical gel; this approach has been
7 particularly explored by Hourdet, Creton and Marcellan (**Figure 2 F**).^{33, 37, 41-45} A
8 striking demonstration of the strength of particle-polymer interactions are 'shake-
9 gels' - solutions of PEO and silica nanoparticles that instantly turn to gels after being
10 shaken or sheared⁴⁶ – or the recent demonstration that silica colloid solutions alone
11 can 'glue' gels or even tissues, due to the adsorption of polymer chains on their
12 surfaces, their reorganisation and capacity to dissipate energy under stress.⁴⁴ The
13 challenge with NC gels based on the adsorption of polymer chains on NPs lies in the
14 adequate dispersion of the particles to avoid flocculation, in the presence of physical
15 forces which can be very high. Indeed, the mixing protocol is critical and different
16 preparation pathways can lead to different outcomes,⁴⁷ while the strength of the
17 interactions can also be used as a parameter to control final structures. NC gels
18 based on physical interactions have been obtained with a wide variety of polymer
19 architectures: block-copolymers,⁴⁸⁻⁵¹ polymers grafted with sticky groups,⁴¹ star
20 polymers⁵² or single-stranded DNA,⁵³ and a range of NPs: laponite,^{49, 54} silica,^{41, 42, 45}
21 cellulose nanocrystals,⁵⁵ graphene oxide sheets,⁵³ metal oxide NPs.⁵¹ In some
22 cases, the interactions, despite being non-covalent, are so strong that the gelation
23 process is virtually irreversible.^{41, 53}

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Based on the results that the adsorption of polymer on NPs alone can lead to a
physical network,⁴¹ Creton et al⁴⁵ have developed networks where the polymer
chains (poly(*N,N*-dimethylacrylamide), PDMA) are slightly crosslinked, thus
combining covalent links (polymer-polymer) to physical links (polymer-NP), an

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3 adaptation of a strategy already used for rubbers.⁵⁶ These hybrid gels show an
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5 increase in compression strength and fracture toughness of notched samples by one
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7 order of magnitude when compared to unmodified PDMA gels,⁴⁵ while the modulus
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9 increases by a factor of 6 with 7% particles. They also exhibit no permanent damage
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11 after several load–unload compression cycles - a very unique property for such
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13 tough hydrogels. The exceptional increase in toughness in the chemically
14
15 crosslinked gels is attributed mainly to the combined effect of breakable silica–
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17 polymer bonds and a wide distribution of elastic chain lengths.⁴⁵ The toughening
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19 mechanisms, similarly to the double-networks developed by Gong,²⁰ are attributed to
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21 a combination of stress redistribution (due to the high functionality of the NP-
22
23 crosslinks) and the existence of dissipative mechanisms slowing down crack tip
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25 propagation (in this case, the break-up/readsorption of PDMA chains to the silica,
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27 which are weaker links than the polymer–polymer bonds). In systems where the
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29 polymer chains do not interact with the silica particles, such as PAAm/silica hydrogel
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31 hybrids, no large increase in strength or modulus is observed, the particles acting as
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33 inert fillers.⁵⁷
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41 **3. Optical properties**

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43 The possibility of exploiting hydrogel/nanoparticle interactions to modulate gels
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45 optical properties was reported by Kirchner and Zsigmondy over a hundred years
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47 ago.¹¹ They discovered that the colour of gelatin hydrogels with embedded gold
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49 nanoparticles would depend on the gel hydration level; the colour change was
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51 ascribed to changes in interparticle distance as the hydrogel network expanded and
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53 contracted due the incorporation or loss of water, showing a very early example of
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55 stimuli-induced optical change in a nanocomposite gel.
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3 A first approach to modify the optical properties of a gel consists in randomly
4 dispersing isotropic NPs within the matrix. One of the most intuitive changes
5 resulting from changes in interparticle spacing is colour - or the absorption of visible
6 light. Several examples can be found in the literature of the modulation of UV-visible
7 absorption⁵⁸⁻⁶² or photoluminescence⁶³⁻⁶⁵ of bulk gel optical properties induced by
8 the incorporation of nanoparticles. For instance, Jeevika and Shankaran developed a
9 colorimetric Cu²⁺-sensor using silver NPs-loaded gelatin hydrogels.⁶⁰ The gelatin
10 matrix offers a stable scaffold to prevent the aggregation of Ag NPs. The sensing
11 mechanism is based on the complexation of Cu²⁺ and the Ag NPs, leading to the
12 agglomeration of the NPs and hence a colour change. A different approach was
13 developed by Qing and co-workers.⁶¹ In this case, functionalized agarose gels were
14 used as templates for the in situ formation of Cu NPs from external Cu²⁺ diffusing
15 into the gel matrix, which, in turn, changed the optical properties of the gel, in this
16 case imparting fluorescence. The group of Caseri reported several studies where
17 they combined polymeric matrices and inorganic nanoparticles (PbS,^{66, 67} FeS⁶⁸
18 and TiO₂⁵⁸) in order to increase the range of refractive indices that could normally be
19 obtained using the polymer gel alone. For a more in-depth discussion of these
20 systems, the reviews of Caseri^{4, 69, 70} are a good starting point.
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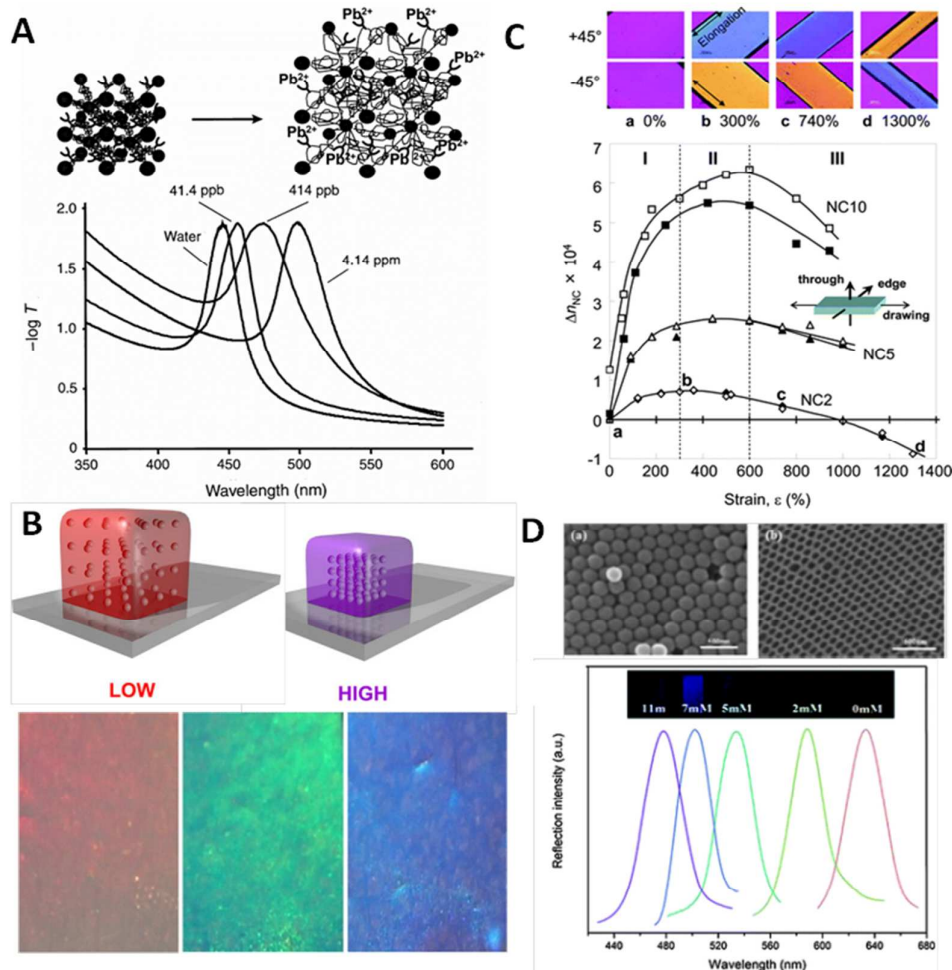


Figure 3. Examples of approaches to impart optical properties to NC gels. **A.** A chemical sensor, capable of detecting different analytes through colour change, was developed based on a crystalline colloidal array of polystyrene (PS) NPs that diffracts light at a wavelength that is determined by lattice spacing. In this example, the visible extinction spectra are shown at various concentrations of $\text{Pb}(\text{CH}_3\text{COO})$ (reprinted by permission from Macmillan Publishers Ltd from reference 71, copyright 1997). **B.** Based on the same principle, photonic crystals made of PS NPs embedded in a PAAm gel were used as sensors of the ionic strength. The periodical arrangement of NPs reflects light, and ions permeating through the gel matrix modify the lattice structure, affecting the wavelength of reflected light. As a result, films of the gel are red when placed in pure water, turn green or purple in the presence of 1 mM or 100 mM electrolyte, respectively, and these can be recorded with a digital camera (reprinted with permission from reference 72. Copyright 2013 American Chemical Society). **C.** The anisotropy of clay NP can be used to impart birefringence - dependent on the amount of clay (NC2-NC10) - when the gel matrix is deformed uniaxially. Photo images show polarized-light micrographs of stretched gels under crossed polarizers (reprinted from

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3 reference 73 with permission from the Royal Society of Chemistry). **D.** Hydrogel films with an
4 inverse opal structure were prepared from SiO₂ colloidal arrays (a) used as a sacrificed
5 template (b) within a gel based on PAAm. The gel responds to varying concentrations of
6 glucose by an increasing blue-shift of the reflection peak (reprinted from reference 74 with
7 permission from the Royal Society of Chemistry).
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11 Changes in optical properties arise from the ability of the gel matrix to support and
12 spatially stabilize NPs, locking the particles in place, in distributions which depart
13 from an isotropic arrangement, thus imparting new properties to the
14 nanocomposites, such as birefringence⁷⁵ or Bragg diffraction (**Figure 3 A**).⁷¹ For
15 instance, starting from an isotropic distribution of the nanoparticles, mechanical
16 deformation of the gel can induce properties not present in the quiescent state, e.g.,
17 strain-induced fluorescence,⁷⁶ or lead to anisotropic properties due to strain-induced
18 organization, for instance, infra-red (IR) dichroism⁷⁷ or optical anisotropy (**Figure 3**
19 **C**).⁷³
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23 A clever example of using a gel matrix to create domain segregation with distinctive
24 optical properties was shown by Raghavan and co-workers,⁷⁸ who developed a
25 hybrid gel made of two components, containing the same monomer but different
26 crosslinkers. Gel 1 contained a chemical crosslinker while gel 2 contained the same
27 monomer, but was physically crosslinked by laponite NPs, inspired from the gels
28 discovered by Haraguchi^{5, 23} described in section 1. The presence of the NPs
29 imparts different properties to the second gel component, such as selectively
30 adsorbing a cationic dye, or birefringence with respect to gel 1, which allowed the
31 researchers to embed a hidden pattern (or 'message') of gel 2 into gel 1, which only
32 becomes visible under cross-polarisers.
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54 Microfluidics have also been used as a tool to generate non-isotropic distributions of
55 NPs in gel matrices. Floyd-Smith and co-workers demonstrated how microfluidic
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3 devices could be used to create non-uniform nanoparticle dispersions in pre-polymer
4 solutions that are preserved upon gelation,⁷⁹ and how this could then be exploited to
5 modulate optical properties, for instance, to create an optical index gradient in
6 polymeric gels loaded with Ti nanoparticles.⁸⁰

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11 Another example of patterned nanoparticles distribution used to impart optical
12 properties of gel nanocomposites arises from the combination of crystalline colloidal
13 arrays (CCA) and polymeric matrices. CCA are self-assembling colloidal particles
14 that form spatially ordered lattices, which show intense Bragg diffraction based on
15 the lattice spacing.⁸¹ It is possible, by embedding the CCA within a gel matrix, to
16 create a nanocomposite with CCA optical properties but also sensitive to changes in
17 the polymeric matrix (**Figure 3 A**).⁷¹ Differently from embedded gold or silver
18 nanoparticles - where the colour changes arise from changes in Plasmon resonance
19 - the optical properties of CCA depend on the lattice spacing, yet, in both cases, it is
20 a change of volume in the polymeric matrix that triggers the optical response. Those
21 systems are particularly useful for diagnostic application, to sense specific analytes,
22 ionic strength, pH, among other properties (**Figure 3 B**).^{71, 72, 82, 83} Another approach
23 is to build so-called 'inverse opal structures', by preparing hydrogels with embedded
24 CCA and then removing the nanoparticles, thus, creating an array of mesopores in
25 the polymeric matrix.⁸⁴ Inverse opal hydrogels offer the advantage of improving the
26 transparency of the hydrogels, at the expense of the cost of removing the
27 nanoparticles. Several examples of tunable inverse opal hydrogels have been
28 described in the literature (**Figure 3 D**).^{74, 85, 86}

4. Biological response

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3 The previous two sections were focused predominantly on the modulation of *physical*
4 properties that can be induced by nanoparticles. However, this is not the only
5 possible outcome of combining NPs and gels: indeed, the biological behaviour of
6 nanocomposites can also be selectively tuned by combining bulk hydrogels and NPs
7 (**Figure 4**).^{2, 8, 13-15, 18} The simplest way to achieve this is by combining materials that
8 already have inherent biological properties into a nanocomposite with desirable
9 physical properties. For instance, the antimicrobial properties of metal NPs is well
10 established.⁸⁷ Therefore, incorporating metal NPs into bulk hydrogels is an easy
11 route towards gels with antimicrobial properties.⁸⁸ For example, Yang and co-
12 workers¹⁶ produced Ag–graphene hydrogels from Ag NPs, graphene oxide and
13 acrylic acid crosslinked by N,N'-methylene bisacrylamide. They demonstrated *in vivo*
14 the hydrogel nanocomposite suitability as skin graft replacement in mice, as well as
15 its antibacterial properties. By combining both polymers of biological origin
16 ('biopolymers') and bio-active NPs, it is possible to obtain so called
17 'bionanocomposites', which have been reviewed in detail elsewhere.^{6, 8, 10, 18, 89} For
18 example, both collagen⁹⁰ and gelatin (denatured collagen)⁹¹ have been combined to
19 Ag NP, with the objective of exploiting gelatin intrinsic biological properties while also
20 imparting antimicrobial activity (**Figure 4 A**). Another layer of complexity can be
21 added to obtain 'smart' bionanocomposites (see also section 5). Leeuwenburgh and
22 co-workers have produced self-healing hydrogels by combining modified hyaluronic
23 acid and calcium phosphate NPs (**Figure 4 B**).⁹² The resultant nanocomposite is
24 particularly suited to bone regeneration as the hyaluronan derivative provides a
25 suitable scaffold for cell growth, the calcium phosphate NPs induce mineralization,
26 while the combination of both in a nanocomposite shows improved mechanical
27 behaviour compared to the blank hyaluronic acid gel, and the self-healing ability
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3 helps preserve the nanocomposite under stressful conditions.⁹² A large range of
4 design options for bionanocomposites is available, and examples can be found
5 elsewhere.^{1, 2, 6, 9, 13, 15, 18}
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10 Nanocomposite hydrogels are particularly well-suited for drug delivery, as drugs can
11 be solubilised into polymeric nanoparticles, which are themselves embedded into the
12 gel matrix, resulting in sustained release.^{2, 6, 13} Shoichet and co-workers have
13 demonstrated the *in vivo* applicability of hyaluronan-based nanocomposites loaded
14 with poly(lactide-co-glycolide) NPs for intrathecal delivery after spinal cord injury in
15 mice.⁹³ However, passive release is not the only biological response that can be
16 obtained. Stimuli responsive nanocomposites are particularly useful for targeted drug
17 delivery;¹⁷ these are covered in the next section. A recent example of how stimuli
18 responsive NC gels can provide unique solutions for engineered drug delivery is
19 given by the 'self-folding nanorobots' developed by Sakar and co-workers, designed
20 for navigation through body orifices and able to release drug on demand (**Figure 4**
21 **C**).⁹⁴ The team of researchers produced a shape-responsive NC gel based on
22 bilayers of poly(ethylene glycol) diacrylate (PEGDA) and PNIPAAm-AAm-PEGDA
23 loaded with either graphene oxide (responsive to near-IR light) or iron oxide NPs
24 (magnetically activated). The shape of the resulting NC gels can be changed on
25 demand and the efficiency of the release of the loaded drugs can, therefore, be
26 modulated.⁹⁴
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47 The object of stimuli-responsive NCs in the biomedical field is not limited to drug
48 delivery. Lu and Zeng produced a nIR stimuli-responsive NC gel based on PNIPAAm
49 loaded with few-walled carbon nanotubes, further coated with a top layer of collagen-
50 functionalized poly(acrylic acid)-co-PNIPAAm in order to improve cell adhesion. This
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NC allowed very precise local mechanical stimuli induced by nIR exposure. The mechanical forces produced can be used to actuate cells (**Figure 4 D**).⁹⁵

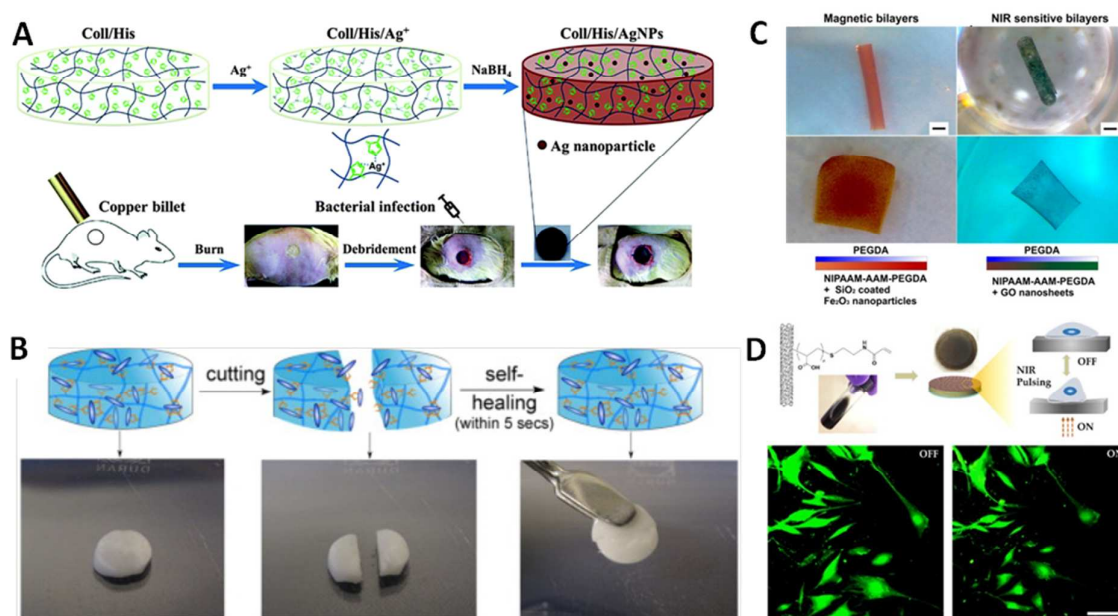


Figure 4. Applications of NC hydrogels to induce a biological response. **A.** Histidine (His) covalently crosslinked to collagen (Coll) scaffolds were employed as a template to chelate silver ions which were reduced in situ to form Ag NPs-hybridized Coll scaffolds. These hybrid materials exhibited enhanced mechanical properties, biocompatibility and antibacterial properties compared to the naked Coll substrate, when used in the regeneration of infected full-thickness burn skin rats (reprinted from reference 90 with permission from the Royal Society of Chemistry) **B.** The synergistic combination of calcium phosphate NPs with a biocompatible matrix of bisphosphate-functionalised hyaluronic acid provides an injectable, yet robust, biodegradable material, displaying self-healing as well as adhesiveness to mineral surfaces. (reprinted from reference 92 ,copyright 2014, with permission from Elsevier) **C.** Self-folding hydrogel bilayers (or ‘microrobots’) that can switch shape - here shown in tubular and rectangular configurations - contain either GO or silica-coated superparamagnetic iron oxide NPs, which provide either nIR responsiveness or magnetic actuation for triggered and targeted drug release, respectively. Scale bar is 500 μm (reprinted with permission from reference 94. Copyright 2015 American Chemical Society). **D.** A NC gel that can deliver spatially and temporally defined mechanical forces to cells was designed by combining uniformly distributed carbon nanotubes to thermally responsive PNIPAAm. nIR stimulation induces changes in strain that can be transmitted remotely to cells and thus offer the

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3 potential to accurately design force sequences for tissue engineering applications (reprinted
4 with permission from reference 95. Copyright 2014 American Chemical Society).
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9 **5. Responsiveness to stimuli**

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11 In the field of soft matter, so-called 'smart' materials have become - over the last
12 decade or so - the object of increasingly intense research. 'Smart' materials are
13 materials that alter their function, properties, shape, etc by responding to specific
14 stimuli.^{1, 96-99} Polymer gels are natural candidates to act as stimuli-responsive
15 materials, since polymer chains can be designed to react to environmental changes,
16 thus altering their solubility, conformation, degree of crosslinking etc, and
17 consequently altering the properties of the gel matrix.^{1, 2, 18, 40, 100, 101} The addition of
18 nanoparticles provides a handle to either impart responsiveness to a 'blank'
19 substrate or alter functionality when embedded in an already 'responsive' gel.^{1, 5} This
20 switchability of stimuli-responsive NC makes them a particular focus of research for
21 controlled drug delivery applications.^{2, 17}
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36 Several strategies have been proposed to impart responsiveness to gel
37 nanocomposites (**Figure 5**). The most obvious approach is to incorporate
38 nanoparticles to improve the applicability of an inherently responsive hydrogel. For
39 this purpose, PNIPAAm and other alkyl-substituted acrylamides have been widely
40 studied, as they undergo a natural thermally-reversible volume transition.^{1, 102}
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42 However, on their own, the mechanical properties of chemically crosslinked
43 PNIPAAm hydrogels are poor, displaying brittleness and low elasticity.⁹⁸ In order to
44 improve their properties, several groups have investigated PNIPAAm gels with
45 embedded NPs (**Figure 5 A**).^{24, 26, 33, 40, 45} Haraguchi^{7, 24, 26, 103} and co-workers
46 produced super-tough alkyl-substituted acrylamide hydrogels embedded with clay
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3 NPs (section 2, **Figure 2 A,B,C**). As already discussed in section 2, the
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5 PNIPAAm/clay nanocomposites showed good transparency, higher malleability and
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7 ductility than the blank hydrogel, whilst preserving the thermal-responsiveness of the
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9 polymer. In a similar vein, Chu and co-workers managed to improve mechanical
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11 performance by crosslinking PNIPAAm in the presence of PNIPAAm-based
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13 nanogels, thus combining thermo-responsive bulk gel and nanogels in a single
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15 system, which resulted in an improvement in response time.⁹⁸ One of the most
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17 interesting aspects of gel nanocomposites is the potential to combine different
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19 elements in order to obtain synergistic improvements. The work of Grunlan and co-
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21 workers⁹⁷ provides a good example; they obtained a double-network (DN) PNIPAAm
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23 hydrogel, where both interpenetrating networks were composed of PNIPAAm with
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25 different crosslink densities, with embedded silica NPs. The manipulation of both
26
27 crosslink densities and NP concentration opened access to a large range of physical
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29 properties, as well as improving the swelling/deswelling kinetics, while preserving the
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31 phase volume transition of PNIPAAm.
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36 Another approach to stimuli responsive gel NCs consists in using nanoparticles to
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38 improve the response, either by increasing its magnitude or coupling it with a
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40 response that is easier to detect. For instance, volume changes are more difficult to
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42 measure than colorimetric changes; therefore, by coupling a visual change to a
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44 volumetric one, the whole process is more readily detected or quantified. In the
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46 optical section, a typical example was presented that involved polymers embedded
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48 with Au or Ag NPs, which colour depends on the interparticle distance: as a result,
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50 gel swelling and deswelling provides an easy handle towards colour changes.^{11, 60}
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54 A more sophisticated strategy is where the NPs themselves are the source of the
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56 stimuli; this is generally obtained by converting an external stimulus to which the
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3 polymer bulk is insensitive (e.g. light) into a stimulus to which the matrix will respond
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5 (e.g. heat). Hilt and co-workers have developed PNIPAAm gels embedded with
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7 coated Fe_2O_3 nanoparticles (**Figure 5 B**).^{6, 104} The iron oxide NPs could be
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9 stimulated by an alternating magnetic field (AMF), which caused them to vibrate in
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11 place and locally generate heat, which, in turn, was able to trigger the expected
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13 phase volume transition of the PNIPAAm bulk matrix; this could be used to induce
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15 the controlled unloading of hydrophobic drugs encapsulated in the gel matrix.¹⁰⁴
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17 West and co-workers have developed a P(NIPAAm-co-AAm) hydrogel with gold
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19 nanoshell NPs for specific drug delivery applications. The gold nanoshells are nIR
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21 absorbers, therefore impacting the polymer phase transition, and enabling optically
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23 triggered drug release.¹⁰⁵ Similarly, Okano and co-workers¹⁰⁶ have used
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25 copolymerization with hydrophobic alkyl methacrylate or hydrophilic acrylamides to
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27 either lower or increase the LCST of the final copolymer. Au NPs efficiently convert
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29 absorbed light to heat, thus generating local hot spots which in turn triggered
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31 P(NIPAAm-co-AAm) transition. A curious example of a response triggered by NPs is
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33 provided by the work of Hayward and his group:¹⁰⁷ by using pattern distribution of
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35 light-absorbing gold NPs in PNIPAAm hydrogel sheets, they were able to induce
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37 local buckling of the sheets and reversibly change the 3D shape of the sheets
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39 (**Figure 5 C**). With a similar approach, Javey and co-workers¹⁰⁸ produced light-
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41 activated actuators of PNIPAAm hydrogels loaded with single-walled carbon
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43 nanotubes (SWNT) combined with low density polyethylene (LDPE). The SWNT can
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45 absorb nIR radiation and dissipate heat, which, in turn, can be used to trigger the
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47 PNIPAAm volume change by nIR exposure. By copolymerizing strips of
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49 PNIPAAm/SWNT on sheets of LDPE, they were able to build large foldable
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51 structures. Graphene oxide (GO) has also been shown to be a good light-sensitive
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3 trigger for PNIPAAm volume transition. Jiang and co-workers¹⁰⁹ produced light-
4 responsive PNIPAAm hydrogels loaded with functionalized GO NPs. The NC
5 showed a good light-triggered volume response, as well as a larger swelling degree
6 than blank PNIPAAm gels; they also demonstrated that the NCs were suitable as
7 microvalves for microfluidic devices.
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10 While PNIPAAm and other alkyl-substituted acrylamide hydrogels are clearly
11 dominating the field of stimuli-responsive NC gels, other responsive polymers have
12 been investigated as gel matrices. Burdick and co-workers¹¹⁰ produced light-
13 sensitive NCs of polypeptides with embedded gold nanorods (**Figure 5 D**). The
14 polypeptide used, PC10P, forms a physical gel at room temperature and melts at ca.
15 60°C. Under nIR exposure, PC10P physical gels loaded with Au nanorods melt at
16 room temperature, due to heat dissipation from the nanorods. Another classic
17 example of thermally responsive polymer is the family of Pluronics, triblock
18 copolymers of PEO-PPO-PEO, which undergo a sol-gel transition in aqueous
19 solution and form micelles into which drugs can be solubilised.^{111, 112} Muhammed
20 and co-workers¹¹³ prepared injectable F127 Pluronic hydrogels loaded with super
21 paramagnetic iron oxide NPs. When submitted to a magnetic field, the NPs re-
22 orientate themselves and cluster together, causing a shrinkage of the bulk gel. This
23 volume contraction accelerated the release of drugs solubilised in the polymer
24 micellar core.
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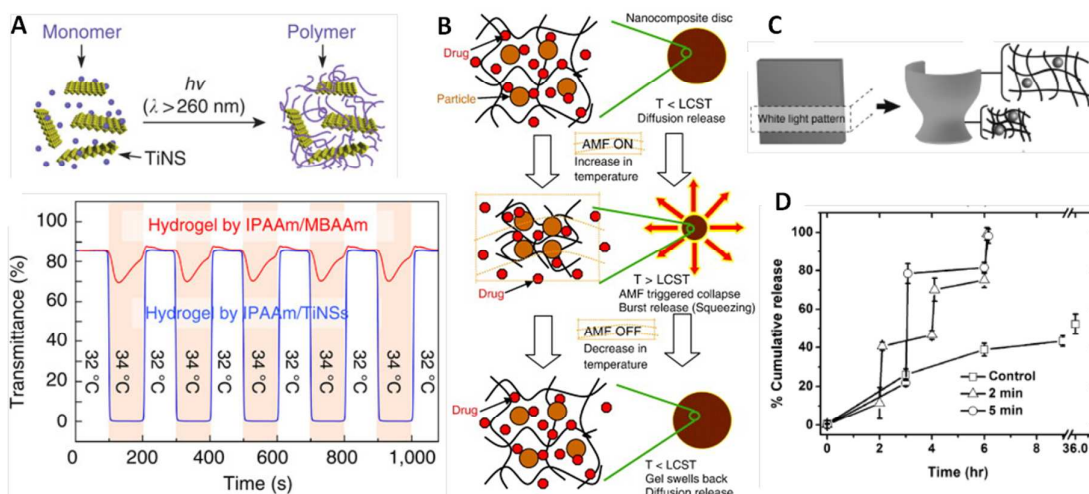


Figure 5. Responsive NC gels. **A.** Schematic structure of titania nanosheet (TiNS)-mediated photoinduced hydrogelation of PNIPAAm ($\lambda > 260$ nm at 25°C for 20 min) and resulting improved temperature-modulated optical transmittance (300 nm) of a 50 μ m hydrogel film (blue) compared to a reference hydrogel with no TiNS (red) (reprinted by permission from Macmillan Publishers Ltd from reference 40, copyright 2013). **B.** Effect of ON-OFF cycles of alternating magnetic field (AMF) on NC gels of PNIPAAm comprising superparamagnetic Fe_3O_4 NPs. AMF triggers uniform heating leading to collapse and resultant burst release of the drug (reprinted from reference 104, copyright 2008, with permission from Elsevier). **C.** Photothermally reprogrammable gels that exploit the thermal deswelling of PNIPAAm networks containing Au NPs, which generate heat when irradiated by light through the SPR effect (reprinted from reference 107 with permission from John Wiley & Sons, Inc.). **D.** Based also on the local heating generated by gold nanorods, a polypeptide gel NC enables step-wise release of FITC-dextran triggered by NIR exposure, compared to a typical diffusion-mediated release when trigger is OFF (reprinted from reference 110 with permission from John Wiley & Sons, Inc.).

6. Structure

While there has clearly been an explosion of new designs for NC gels, either for the creation of transparent, ultra-tough, super extensible gels, or to impart responsiveness, mainly with biomedical applications in mind, the understanding of the intricate structural arrangements that lead to these superior properties is still lagging behind. Small-angle X-ray and neutron scattering (SAS) are unique

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3 techniques that can probe gel morphology^{114, 115} and nanoparticle arrangement, and
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5 probably the only tools to achieve nanoscale level information on nanocomposite
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7 structure.¹¹⁶⁻¹¹⁹ This final section summarises information obtained on selected NC
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9 gels, some which have been described above. There are clear difficulties with
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11 obtaining structural information, the most obvious one being to deconvolute
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13 contributions arising from the particles and the gel matrix.
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17 Most of the structural work on NC gels has been published on thermally-responsive
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19 block-copolymers,^{48, 49, 51, 117, 118} such as Pluronics, whose gelation occurs through
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21 the arrangement of micelles into a macro-lattice.¹²⁰ SANS can therefore be used to
22
23 assess whether the presence of NPs disrupts this arrangement (even if only
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25 qualitatively), and - if sufficient contrast is available between the polymer and the
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27 NPs - provide some insight into the localisation of the NPs within this highly ordered
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29 structure.
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33 Ogata and co-workers¹¹⁸ have developed organic-inorganic nanocomposite gels as
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35 an in situ gelling biomaterial for injectable accommodative intraocular lens, based on
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37 a hydrophobically modified PEG containing hydrophilized silica. SANS was used to
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39 ascertain that the incorporation of the NPs (2-5 nm) did not disrupt the micellar
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41 ordered structure at the origin of the gel phase.¹¹⁸ Instead, when Fe₃O₄ particles
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43 were introduced into a closed-packed network of F108 Pluronic micelles (25%),⁵¹ it
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45 was found that the NPs (of size comparable to the micelles) disrupted the
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47 macrocrystalline order, resulting in the clustering of the nanoparticles.^{48, 51}
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51 Raghavan and co-workers have used SANS to help elucidate the mechanism of
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53 gelation in mixtures of Pluronic and laponite, either thermally induced⁴⁹ or by
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55 lowering the pH, through the UV-triggered activation of a photoacid generator,¹¹⁷ at
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57 polymer concentrations well below the close-packing of micelles (**Figure 6**). SANS
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data show that at either low temperature or high pH the laponite particles are dispersed in solution, stabilised by Pluronic chains adsorbed through its hydrophobic segments. Upon increasing the temperature or reducing the pH, the polymer desorbs and form micelles in solution, which drive the clustering of laponite into a volume-filling network⁴⁹ or house-of-card structure (Figure 6).¹¹⁷

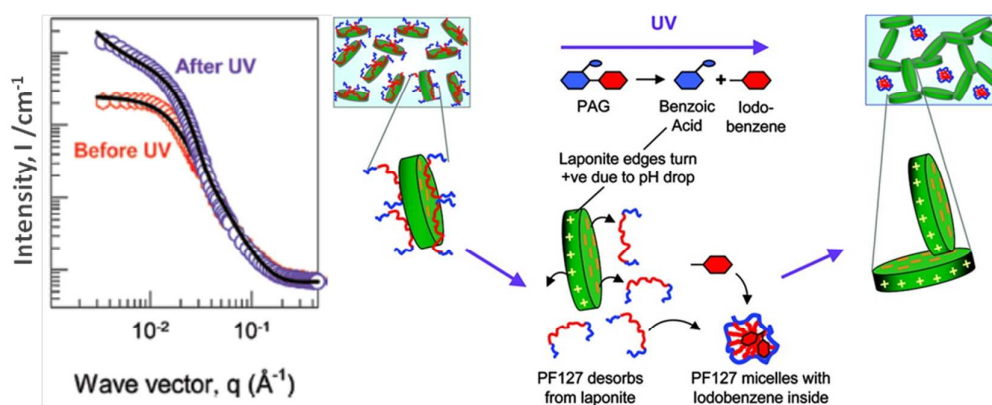


Figure 6. Scheme representing the photogelling mechanism in mixtures of F127 Pluronic, laponite and a photoacid generator (PAG). Initially, laponite NPs are stabilised by the polymer, with the hydrophobic PPO (red) segments adsorbed on their surface and the SANS data (left) can be fitted by a disk-shape with an adsorbed polymer shell. Upon UV irradiation of the PAG, the pH drops, laponite edges become positively charged, F127 desorbs and forms micelles in solution while interactions between laponite NPs result in a house-of-cards structure, which appears as an increase in the scattered intensity at low- q , corresponding to a fractal dimension of 2 (reprinted with permission from reference 117. Copyright 2009 American Chemical Society).

Still in mixtures of laponite with another Pluronic (P123), but over a much higher concentration regime (50%), SAXS data have been used to show that the addition of laponite drives a transition from a hexagonal phase of rodlike micelles at low temperature to a lamellar phase at high temperature.⁵⁴

SANS offer the advantage over X-rays (SAXS) to contrast-match specific parts of the system by selective deuteration, or by simple tuning of solvent composition (H_2O/D_2O mixture). This is particularly applicable where the NPs are inorganic and

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3 thus have a neutron scattering length density quite different from the polymer, thus
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5 their organisation within the gel matrix can be visualised with the polymer made
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7 'invisible'. Marcellan and co-workers³⁷ thus were able to ascertain that silica NPs –
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9 acting as multifunctional crosslinks - were well dispersed in a PDMA matrix - even
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11 after cyclic mechanical loading. Another benefit of SAS techniques is the possibility
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13 to visualise structural changes while stretching a material. With the system just
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15 mentioned,³⁷ it was possible to show that under an applied strain the silica particles
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17 arranged themselves perpendicular to the stretching direction, but the isotropic
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19 pattern was recovered when at rest.
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23 A large amount of structural work was performed by the group of Shibayama^{22, 119,}
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25 ¹²¹⁻¹²⁵ to elucidate the origin of the exceptional mechanical properties of the
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27 clay/polymer nanocomposite gels developed by Haraguchi and described in section
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29 ^{25, 23, 28, 121} (**Figure 7 A,B**), using a combination of SANS, SAXS and light scattering.
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31 With SANS and contrast variation on stretched NC gels (**Figure 7**), an abnormal-
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33 butterfly pattern was observed, which is commonly seen in chemical gels under
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35 strain and assigned to inhomogeneities, but was here assigned to the orientation of
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37 the clay platelets with their surface parallel to the stretching direction (**Figure 7**
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39 **C,D**).¹²² These structural studies thus made it possible to attribute the remarkable
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41 extensibility and strength at break of these pioneering NC gels to the following
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43 factors: a very large inter-crosslink distance (which is set by the inter-platelet
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45 distance); the length of the topological chains $\langle R^2 \rangle^{1/2}$, where R is the end-to-end
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47 distance in PNIPAAm chains between crosslinks; and, finally, the number of
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49 'effective' crosslinks' - much larger than in conventional gels -, because most
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51 polymer chains are anchored to clay platelets and therefore elastically active (**Figure**
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53 **7**).
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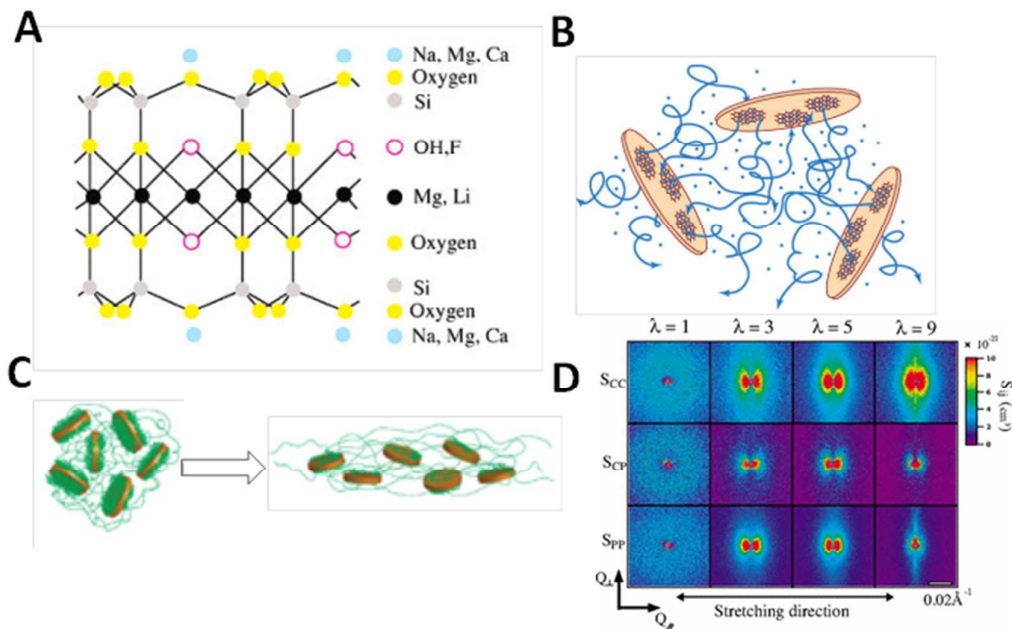


Figure 7. Schematic representing the structure of the clay/PNIPAAm gels. **A.** Chemical structure of the clay platelets showing the various types of atoms. The oxygen atoms may form hydrogen bonds with the carbonyl oxygen of NIPAAm monomer, binding them to the surface, (reprinted with permission from reference 122. Copyright 2005 American Chemical Society) **B.** hence PNIPAAm is polymerised from the surface thus creating bridging chains between the platelets. (reprinted with permission from reference 122. Copyright 2005 American Chemical Society) **C.** Upon stretching, the platelets orient with their surface parallel to the strain direction – as established by (reprinted by permission from Macmillan Publishers Ltd from reference 126. Copyright 2011) **D.** the two-dimensional SANS partial structure factors of the NC gels obtained by systematic contrast variation (S_{CC} : clay scattering; S_{CP} : clay polymer cross term; S_{PP} : polymer scattering) at varying stretching ratios λ . The remarkable mechanical properties of these NC gels are thus assigned to the length of the chains between the platelets, which act as multifunctional crosslinks, and the high density of elastically active polymer chains (reprinted with permission from reference 125. Copyright 2009 American Chemical Society).

7. Conclusions

This review has demonstrated the huge variety of structures, designs, properties that can arise from the combination of nanoparticles with hydrogels. Nanocomposite gels provide a relatively simple and flexible concept to generate specific properties and modulate them by varying composition, often circumventing the need of complex chemistry. It is noticeable that considerable effort has been targeted towards biomedical materials, where the 'softness' and hydrated environment offered by hydrogels provide a mimic of natural tissues, which, in turn, can be modulated by nanoparticles for improved robustness, elasticity, biocompatibility, or to add features such as antimicrobial, self-healing, controlled release, etc. A very strong focus of soft matter research has been in the design of 'smart' materials: materials which alter their behaviour in response to a trigger. Evidently, nanocomposite gels offer an ideal platform to do just this: synergistic interactions between the gel matrix and the nanoparticle dispersion offer a formidable handle to impart responsiveness to environmental factors such as temperature, pH, light, strain – as demonstrated in this review. As a result, a whole new generation of soft nanocomposites promises to find useful applications as tools for diagnostic and as scaffold biomaterials for 3D cell culture, which can guide cell behaviour through a variety of physical cues. Overall, the trend is towards increasingly multi-functional, complex materials which can modulate their behaviour as a response to several, rather than one, stimuli. It is however important to emphasize that the combination of simple materials alone can lead to very interesting outcomes, as demonstrated recently with silica sols acting as strong adhesives for gels and biological tissues,⁴⁴ which is simply due to the dynamic adsorption of multiple polymer chains on the nanoparticles surface. In addition, in

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3 this race towards ever-more sophisticated, cutting-edge materials, one should not
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5 forget the necessity of fundamental mechanistic and structural understanding.
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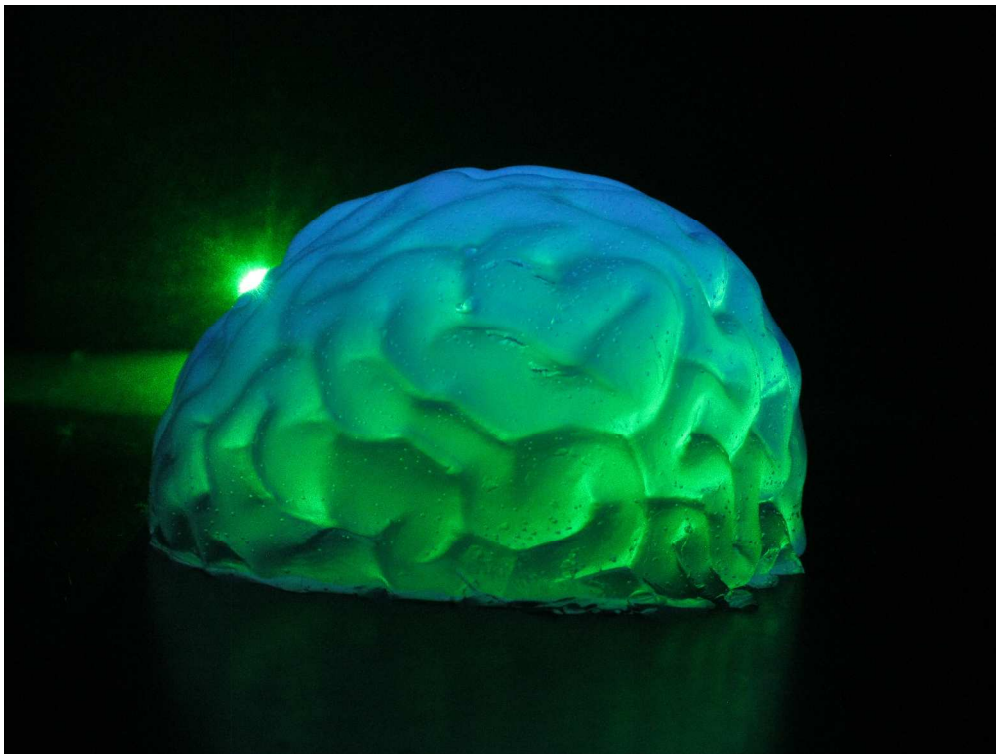
'Soft' nanocomposites: nanoparticles to tune gel properties

Marcelo A. da Silva, Cécile A. Dreiss*

Gel nanocomposites offer a simple yet powerful concept towards material design, with the combination of nanoparticles functionality (here: luminescence) with the 'soft' properties of a gel matrix (here: shape retention and mechanical support).

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Review

Soft nanocomposites: nanoparticles to tune gel properties

Authors' biographies



Cécile A. Dreiss obtained her PhD in Chemical Engineering from Imperial College London, followed by a postdoc in colloid science at the University of Bristol. Now based at King's College London, her research focuses on understanding and exploiting self-assembly in soft matter, spanning colloidal, polymeric and biological systems, by establishing relationships between properties on the macro-scale (such as rheology) and the organization on the nanoscale, using mainly techniques such as small-angle neutron scattering.



Marcelo A. da Silva obtained his PhD in physical-chemistry from the State University of São Paulo, Brazil, with a project on solvent-induced protein gelation. After his PhD, he worked as a postdoctoral researcher on surfactants as drag-reduction additives at the State University of Campinas. He then took up a postdoctoral position at King's College London working on biopolymer gels for tissue engineering applications, and is currently a research associate at the University of Leeds studying hydrogels of polyproteins.